

## The Crystal Structure of Basic Tellurium Nitrate, $\text{Te}_2\text{O}_4 \cdot \text{HNO}_3$

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(Received 24 January 1966)

The structure of  $\text{Te}_2\text{O}_4 \cdot \text{HNO}_3$  (the product of dissolving Te or  $\text{TeO}_2$  in  $\text{HNO}_3$ ) has been determined from visual intensities collected with Mo  $K\alpha$  radiation, and refined to  $R=7\%$ . The crystals are orthorhombic,  $Pnma$ , with 4 formula units in a unit cell of dimensions  $a=14.55$ ,  $b=8.77$ ,  $c=4.45$  Å. The structure consists of puckered layers of tellurium and oxygen atoms. Each tellurium atom is linked to one other by two oxygen bridges and to two others by a single oxygen bridge each. The four tellurium-oxygen bonds around a tellurium atom are directed approximately toward the axial and two equatorial apices of a trigonal bipyramid. The tellurium-oxygen bond lengths range from 1.88 to 2.16 Å, with an average estimated standard deviation of 0.015 Å. The nitrate group is hydrogen-bonded to one of the bridging oxygen atoms, with  $\text{O} \cdots \text{O}=2.75$  Å.

### Introduction

Although tellurium exhibits a variety of coordination patterns, accurate crystal structure details are rare even for its simple compounds. For instance, the two crystallographic forms of  $\text{TeO}_2$  have been studied in some detail: tellurite (orthorhombic  $\beta\text{-TeO}_2$ ) has been classified as a distorted brookite-type structure (Ito & Sawada, 1939); tetragonal  $\alpha\text{-TeO}_2$  gave fair agreement as a direct analogue of rutile (Goldschmidt, 1926), but more recent studies (Stehlik & Balak, 1949; Leciejewicz, 1961) show that it is a distorted variant of the rutile structure. Except for values in the last of these references, realistically estimated standard deviations for tellurium-oxygen bond lengths are greater than 0.1 Å. The structure of 'basic tellurium nitrate',  $\text{Te}_2\text{O}_4 \cdot \text{HNO}_3$ , was selected for study not only to determine the nature of this particular compound, but also to provide more information about coordination patterns around tellurium and about tellurium-oxygen bond lengths.

### Experimental

A preliminary investigation of this material (Heiart, 1956) yielded the following crystallographic data:

Orthorhombic	$a = 14.547$ (0.042) Å
$\rho_{\text{obs}} = 4.36$ g.cm <sup>-3</sup>	$b = 8.770$ (0.006)
$\rho_{\text{calc}} = 4.48$ g.cm <sup>-3</sup> for $Z=4$	$c = 4.446$ (0.004)

where values in parentheses are estimated standard deviations. Systematic absences indicate that the space group is  $Pnma$  or  $Pn2_1a$ .

Basic tellurium nitrate was readily prepared by dissolving metallic tellurium in nitric acid, followed by several recrystallizations. A small single crystal was selected and used to obtain precession  $h0l$ ,  $h1l$ ,  $0kl$ ,  $lkl$  data, then Weissenberg  $hkl$  data for  $l=0$  through

6 with the usual multiple film technique; Mo  $K\alpha$  radiation was used to reduce absorption further. Intensities were estimated visually on the basis of a series of standard intensity scales. Lorentz-polarization corrections were estimated graphically for the precession data and the Zalkin-Templeton Incor program was used to correct the Weissenberg data.

For further experimental evidence relating to hydrogen bonding, infrared spectra were obtained by both KBr pellet and Nujol mull techniques. The instrument used was a Perkin-Elmer 21 and the pellet spectrum was calibrated against that of polystyrene.

### Structure determination

In view of the short  $c$  axis and the presence of a heavy atom, it was originally expected that the structure could be solved in projection. A Patterson projection down the short  $c$  axis located the tellurium atoms (Heiart, 1956). This same projection,  $P(u,v)$ , showed a row of prominent peaks on the line  $P(0,v)$  which strongly suggested atoms related by (010) mirror planes and thus indicated  $Pnma$  as the probable space group rather than  $Pn2_1a$ . The projection did not lead to the complete structure because what seemed to be too many peaks were found in the electron density calculated with signs determined by the tellurium atom (the correct structure contains only two independent oxygen atoms in eightfold general positions and three independent oxygen atoms in fourfold special positions instead of four and one, as originally guessed).

Subsequent work used three-dimensional calculations. The  $z$  coordinate of the tellurium atom was estimated to be near zero from the generally small intensities of reflections depending on the factor  $\sin 2\pi lz$ , and in fact to be roughly 0.02 from the slightly faster-than-normal decline with increasing  $l$  for reflections depending on the factor  $\cos 2\pi lz$ .

Signs were allotted to all reflections according to the tellurium contribution, and the three-dimensional difference density was calculated with tellurium contri-

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butions subtracted. The difference map indicated considerable anisotropy in the thermal motion of all atoms and was especially rough in the tellurium neighborhood. Nevertheless, the correct number of peaks were found in structurally reasonable positions and starting parameters were assigned to all atoms except the hydrogen atom.

Refinement was continued using the BXLS block diagonal least-squares program (Carpenter, 1963) minimizing the quantity

$$R_w = [\sum w(\Delta F)^2 / \sum w|F|_o^2]^{\frac{1}{2}}$$

with a single scale factor for all data. The weights,  $w$ , were assigned according to the formulas:

$$w = \sigma^{-2}, \quad \sigma \approx 2.3 + 0.08N^{-\frac{1}{2}}|F|_o + 0.005|F|_o^2,$$

$N$  being the number of independent observations for a particular structure factor. The coefficients, originally estimated from the internal consistency of the data, are expressed above on the same scale as the values of  $|F|_o$  in Table 2. Reflections too weak for accurate measurement were given zero weight.

Five cycles of isotropic refinement reduced the discrepancy indices from  $R=0.178$  and  $R_w=0.263$  to  $R=0.114$  and  $R_w=0.157$  for observed reflections only. After the fifth cycle, a difference map indicated that the prominent remaining discrepancy was due to anisotropic thermal motion.

Because of the overwhelming contribution from tellurium to most of the structure factors, it was expected, and later confirmed, that anisotropic thermal parameters for the light atoms would not refine properly. Several cycles of partial anisotropic refinement were carried out by manually shifting to equivalent isotropic parameters for the lighter atoms after each cycle. This refinement procedure converged to a structure with  $R=0.088$ ,  $R_w=0.114$ .

Since the difference map was still rather rough near the tellurium atom, further refinements were necessary. Before proceeding, extinction corrections were applied to reflections corresponding to the 25 strongest observed intensities using the approximation

$$F_{\text{corr}} = F_{\text{obs}}(1 + cJ_{\text{obs}})$$

(Zachariasen, 1963) and the recent HFS scattering factor (Hanson, Herman, Lea & Skillman, 1964) for tellurium was corrected for the real part of the anomalous dispersion (Cromer, 1965) for use in subsequent refinement. Lorentz-polarization corrections and inter-layer scale factors had been applied to the original data by hand and a reflection-by-reflection check of the data uncovered a number of errors. The effect of extinction on the  $hk0$  reflections and a systematic error in  $L_p$  corrections for  $hk6$  data required that the observed structure factors for these layers be scaled down by 5%.

Concurrently, the BXLS program was modified to allow refinement of some atoms while including others only for their contribution to the structure factors. This made it possible to run alternate cycles, refining the tellurium anisotropically while holding the lighter atoms fixed, then refining the light atoms isotropically while holding the tellurium parameters constant.

Refinement was continued until all coordinate shifts were  $<10\%$  of their calculated standard deviations. Final discrepancy indices for the 426 observed reflections only were  $R=0.069$  and  $R_w=0.092$  with final atomic coordinates of Table 1. Table 2 lists the observed and calculated structure factors based on the final parameters given in Table 1.

The final difference function was within the range of expected fluctuations except for two peaks associated with a periodic disturbance (period  $c/6$ ) extending in the  $c$  direction from the tellurium atom. The largest such peak had a height of  $4 \text{ e.}\text{\AA}^{-3}$  ( $2.8\sigma$ ), less than half the height of the smallest oxygen peak in the first difference map.

### Discussion of the structure

The overall packing plan for basic tellurium nitrate is a series of puckered tellurium-oxygen networks extended in two dimensions normal to the  $a$  axis. The nitrate groups are connected to this framework by

Table 1. *Final atomic parameters*  
Estimated standard deviations are in parentheses\*

	$x/a$	$y/b$	$z/c$	$B$
Te(1)	0.16655 (10)	0.06355 (17)	0.02001 (24)	†
O(2)	0.2106 (11)	0.0279 (19)	0.4110 (36)	0.98 (21) $\text{\AA}^2$
O(3)	0.2334 (16)	$\frac{1}{4}$	0.9234 (52)	1.00 (33)
O(4)	0.0917 (20)	$\frac{1}{4}$	0.2241 (66)	1.88 (46)
N(5)	0.4386 (23)	$\frac{1}{4}$	0.7105 (70)	1.65 (48)
O(6)	0.3787 (18)	$\frac{1}{4}$	0.5267 (61)	1.52 (38)
O(7)	0.4662 (16)	0.1260 (26)	0.8260 (53)	2.37 (37)
Te(1)	$\beta_{11}=0.00152$ (4)	$\beta_{22}=0.00271$ (11)	$\beta_{33}=0.00462$ (32)	
	$\beta_{12}=0.00013$ (21)	$\beta_{23}=0.00008$ (65)	$\beta_{31}=0.00030$ (39)	
for a temperature factor of the form				
$\exp [-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + \beta_{12}hk + \beta_{23}kl + \beta_{31}lh)]$				

\* Thermal parameters from a block diagonal refinement and standard deviations from a diagonal approximation, as given in this table, are less meaningful than those given by a full matrix refinement.

† Equivalent isotropic parameter (for comparison)  $\hat{B}_{\text{Te}}=0.83$ .

hydrogen bonds and all distances between atoms in adjacent layers are larger than the normal van der Waals distances. Therefore, details of the arrangement of atoms within one of these layers is of primary interest.

Fig. 1 shows a section of this network projected along the *a* axis. The principal interatomic distances and angles are listed in Table 3. Fig. 2 is a *b*-axis projection of the section between adjacent mirror planes. A stereoscopic drawing, Fig. 3, is included for those who have stereo viewers or flexible eyes.

The curious tellurium-oxygen bonding pattern can be thought of as the skeleton of the structure. Distorted double oxygen bridges join tellurium atoms across the mirror planes and single oxygen bridges between tellurium atoms link these Te<sub>2</sub>O<sub>2</sub> units into a network normal to the *a* axis. Around any one tellurium atom there are four short tellurium-oxygen distances between 1.88 and 2.16 Å, and three longer distances between 2.8 and 2.9 Å. The difference between any two of the four shortest distances is definitely sig-

nificant (confidence level  $\geq 0.999$ , based on the BXLS formal estimated standard deviations). The shortest tellurium-tellurium distance is 3.27 Å across the mirror plane, and successive tellurium atoms in the -Te-O-Te- chain are 3.48 Å apart.

Since the tellurium-oxygen bonding in Te<sub>2</sub>O<sub>4</sub>.HNO<sub>3</sub> may be expected to be closely related to that in tellurite ( $\beta$ -TeO<sub>2</sub>) and tetragonal  $\alpha$ -TeO<sub>2</sub>, the results reported for these compounds were examined in detail. The data for tellurite (Ito & Sawada, 1939), were found to be inadequate for extraction of sufficiently precise structural detail (we plan further work on tellurite).\* In the most recent report on the structure of  $\alpha$ -TeO<sub>2</sub> (Leciejewicz, 1961) several errors (interatomic distances, angles, etc.) were found although the atomic coordinates seem to be correct. Allowing for these discrepancies, a similar arrangement for the four nearest neighbor oxygen atoms about the tellurium

\* Since the completion of the manuscript of this paper, an accurate redetermination of the structure of tellurite (Beyer, Sahl & Zemann, 1965) has come to our attention.

Table 2. Observed and calculated structure factors for observed reflections ( $\times 10$ ). Estimated upper bounds and calculated structure factors for reflections too weak to measure ( $\times 10$ )

H	K	FOBS	FCALC	H	K	FOBS	FCALC	H	K	FOBS	FCALC	H	K	FOBS	FCALC	H	K	FOBS	FCALC	H	K	FOBS	FCALC
L=0				4	8	929	967	7	3	1948	-2076	10	6	598	602	1	11	861	858	1	1	550	-438
				6	8	1722	-1841	11	3	1291	1490	12	6	989	-945	1	2			1	2	536	431
2	0	1510	-1911	8	8	823	805	13	3	1246	-1315	2	7	667	-649	L=4				2	2	525	-557
4	0	1480	-1335	12	8	745	681	0	5	549	-521	4	7	608	593					3	2	1027	-1018
6	0	2779	2665	14	8	473	497	5	5	1850	1827	1	8	1843	-1745	0	0	1767	1850	4	2	682	667
8	0	1352	-1369	16	8	394	459	7	5	1608	-1597	2	8	701	687	2	0	808	-791	4	2	778	621
10	0	1074	-1084	18	8	929	-800	11	5	1397	1527	4	8	745	797	4	0	1012	-1095	10	2	578	435
12	0	1639	1637	20	8	416	-475	13	5	1216	1266	6	8	1571	-1506	6	0	906	910	0	3	1042	-1025
14	0	846	-777	2	9	694	626	1	6	1095	-1009	8	8	929	692	8	0	1563	1618	1	3	1238	-1203
16	0	823	-822	4	9	373	-411	3	6	1805	1818	12	8	1284	-1165	7	0	808	-870	2	3	476	476
18	0	1110	1146	6	9	778	765	5	6	753	-819	0	10	747	-898	8	0	720	-715	5	3	929	937
20	0	498	-485	8	9	587	-521	7	6	623	-617	2	11	944	912	10	0	702	-717	6	3	762	-820
22	0	408	-423	14	9	301	286	9	6	1133	1138					11	0	588	572	7	3	1065	-965
24	0	652	670	16	9	315	-375	5	7	730	754	L=5				12	0	1193	1286	11	3	823	769
4	1	929	-1027	0	10	1178	-1207	1	8	989	-907					13	0	564	-634	0	5	750	-735
6	1	1646	1653	2	10	559	506	3	8	1616	1714	1	0	1455	1395	14	0	559	-554	1	5	1049	-975
8	1	437	-492	4	10	494	486	5	8	846	-927	2	0	936	-811	16	0	413	-409	5	5	1027	949
10	1	853	868	6	10	793	-740	7	8	830	-815	4	0	1846	-2266	3	1	287	358	6	5	913	-744
14	1	725	-771	8	10	279	373	9	8	1442	1530	4	0	1057	1073	4	1	472	-549	7	5	921	-847
16	1	516	452	12	10	762	-660	1	9	808	707	5	0	1269	1194	8	1	447	-532	1	5	582	-474
0	2	1911	2033	18	10	438	-474	1	10	631	-528	7	0	1117	1114	9	1	385	429	3	6	746	743
2	2	1812	-1703	2	11	1057	1020	3	10	891	969	8	0	574	-588	10	1	637	509	9	6	726	744
4	2	868	-758	4	11	1004	-964	1	11	1034	1142	4	0	1558	-1740	2	2	1034	1062	0	7	393	-311
6	2	2500	2463	8	11	1034	970					10	0	649	683	2	2	575	-614	3	8	929	92
8	2	1125	-1111	10	11	861	-856	L=2				11	0	654	579	4	2	429	457				
10	2	793	-603	14	11	883	792	0	0	2870	2959	15	0	1231	-1228	5	2	429	457				
12	2	1178	1182	16	11	724	-655	2	0	2039	-2068	17	0	1709	-6308	6	2	1231	1189	L=6			
14	2	511	-400	20	11	409	338	0	0	326	-378	1	1	725	-711	8	2	484	-477				
16	2	702	-593	2	13	823	755	3	0	1518	-1404	5	1	669	735	10	2	504	-568	0	0	921	981
18	2	868	755	4	13	793	-700	4	0	981	967	6	1	436	-450	11	2	506	564	1	0	921	-917
20	2	301	-253	8	13	823	750	5	0	2417	2592	7	1	707	-665	12	2	725	906	2	0	311	-370
22	2	315	-335	10	13	778	-677	6	0	906	910	11	1	637	662	1	3	324	-388	4	0	574	-585
24	2	437	499	0	14	416	544	8	0	906	910	11	1	468	-408	2	3	1420	-1440	5	0	732	813
2	3	2266	-2727	6	14	778	700	10	0	1012	-926	13	1	929	836	3	3	883	962	6	0	1087	1041
4	3	2689	2707	12	14	516	547	11	0	573	607	1	2	785	-750	4	3	1359	1297	7	0	717	-747
6	3	285	159	0	16	531	723	12	0	442	-428	3	2	1472	-1502	7	3	399	-420	8	0	416	-421
8	3	2031	-1890	6	16	531	608	13	0	701	643	4	2	616	623	8	3	1102	-1239	11	0	598	636
10	3	1812	1742					14	0	589	-582	5	2	702	748	9	3	513	610	12	0	959	773
14	3	1042	-990	L=1				16	0	1148	1076	7	2	475	572	10	3	497	496	4	1	337	346
16	3	1012	950	1	0	1646	1788	1	1	326	-257	9	2	1253	-1288	14	3	700	-790	0	2	580	690
20	3	921	-885	0	0	1208	-1117	2	1	785	-862	15	2	887	-793	2	5	1072	-1156	1	2	580	-575
22	3	524	622	3	0	3361	-3554	4	1	815	905	0	3	1208	-1135	3	5	518	502	2	2	339	-321
4	4	583	580	5	0	1004	1000	6	1	441	-401	1	3	1767	-1820	4	5	1163	1130	4	2	344	-356
2	4	481	-528	5	0	985	-473	8	1	861	-898	3	5	653	596	8	5	959	-1057	5	2	543	555
4	4	186	-140	6	0	595	-473	10	1	555	633	5	5	1812	1821	9	5	651	788	6	2	605	630
6	4	574	-498	7	0	1639	1606	10	1	457	-502	6	3	616	-608	10	5	981	1014	7	2	540	-549
10	4	258	209	8	0	376	-283	14	4	457	-502	7	3	1420	-1322	14	5	734	-796	12	2	626	613
2	5	2094	-2243	9	0	2243	-2345	16	1	478	435	7	3	1420	-1322	14	5	734	-796	12	2	626	613
4	5	2099	2122	11	0	997	1009	0	2	2001	2168	11	3	1216	1169	6	6	1160	-1186	1	3	543	-481
6	5	459	-405	13	0	936	962	4	2	966	-895	12	3	546	-477	1	5	580	493	2	3	876	-820
8	5	1503	-1492	15	0	1442	-1506	6	2	1639	1606	13	3	1042	-1036	4	6	580	571	3	3	868	878
10	5	1321	1291	17	0	883	794	8	2	1012	-1013	17	3	793	757	5	6	511	-561	4	3	785	777
14	5	1367	-1245	0	1	299	230	10	2	688	-596	0	5	707	-677	6	6	861	-931	5	3	457	-432
16	5	929	890	1	1	1344	-1225	12	2	1178	1190	1	5	1586	-1523	12	6	629	693	8	3	755	636
20	5	738	-597	2	1	345	349	1	3	567	-506	5	5	1420	1266	0	8	1170	-1188	4	5	717	755
22	5	487	553	3	1	220	244	4	3	2522	-2401	6	5	648	-700	1	8	685	636	10	3	660	641
0	6	2016	-2109	4	1	491	443	3	3	491	452	7	5	1480	-1451	2	8	620	639	2	5	793	-684
2	6	815	784	5	1	672	689	4	3	2039	1969	11	5	1027	1085	4	8	555	517	3	5	883	793
4	6	951	975	7	1	947	-987	8	3	1729	-1662	1	8	135	-673	4	0	815	-803	4	5	755	636
6	6	1306	-1337	11	1	731	775	10	3	1460	1387	1	6	669	-634	0	8	1125	-1150	8	5	717	755
8	6	695	604	13	1	685	-689	14	3	1133	-1101	3	6	1442	1325	12	8	913	-861	9	5	676	653
10	6	752	696	1	2	913	823	16	3	981	894	5	6	580	-546					10	5	717	631
12	6	1238	-1169	2	2	1686	-1815	2	5	1827	-1680	7	6	637	-619	L=5				1	6	645	-650
14	6	537	528	4	2	722	761	3	5	563	475	9	6	1027	773					1	6		



lurium–oxygen bonds may be rationalized in the following way. In the first place, the fact that the axial bonds are longer than the equatorial bonds in both  $\text{Te}_2\text{O}_4 \cdot \text{HNO}_3$  and  $\alpha\text{-TeO}_2$  can be interpreted in terms of different hybrids used by tellurium for the two kinds of bond; e.g.  $5s\ 5p_x\ 5p_y$  for equatorial bonds,  $6s\ 5p_z$  for axial bonds, would lead to the observed difference. In the second place, tellurium–oxygen bonds within the four-membered  $\text{Te}_2\text{O}_2$  rings in  $\text{Te}_2\text{O}_4 \cdot \text{HNO}_3$  are longer than those outside, consistent with a postulated strain in the rings. Further, the strain in a particular part of either structure seems to be measured by the deviation of the tellurium–oxygen–tellurium angle from an ideal value, apparently near  $126^\circ$ . The effect of the strain on the bond lengths is slightly less than the effect of hybridization.

The intermediate-length tellurium–oxygen distances could suggest weak bonding, but the lack of some coherent pattern in the different compounds makes this approach uncertain. Alternatively, the van der Waals radius of a tellurium atom bonded to four more electronegative atoms should be considerably less than the usual  $2.2\ \text{\AA}$  value based on the ionic radius of  $\text{Te}^{2-}$  (Pauling, 1960). A value of about  $1.74\ \text{\AA}$  is consistent with the experimental distances, and then no tellurium–tellurium bonding need be postulated.

The distance between O(6) and O(3) is  $2.75\ \text{\AA}$  and the angle  $\text{N-O(6)}\cdots\text{O(3)}$  is  $97^\circ$ , indicating a fairly strong hydrogen bond,  $\text{O(6)-H}\cdots\text{O(3)}$ . The infrared spectrum confirms the presence of hydrogen bonding. In each spectrum a rather broad (half width  $\approx 300\text{cm}^{-1}$ ) and intense ( $\approx 40\%$  absorption) peak centered near

$250\text{cm}^{-1}$  was observed. For an interaction of the type  $\text{O-H}\cdots\text{O}$ , this would imply an  $\text{O-H}\cdots\text{O}$  distance of approximately  $2.74\text{--}2.76\ \text{\AA}$  (Nakamoto, Margoshes & Rundle, 1955).

The authors wish to express their thanks to Mr B. Annis, Mr L. A. Chasin, and Mr W. S. Magee (former students at Brown) for collecting the data and for preliminary calculations, to Dr C. K. Johnson (Oak

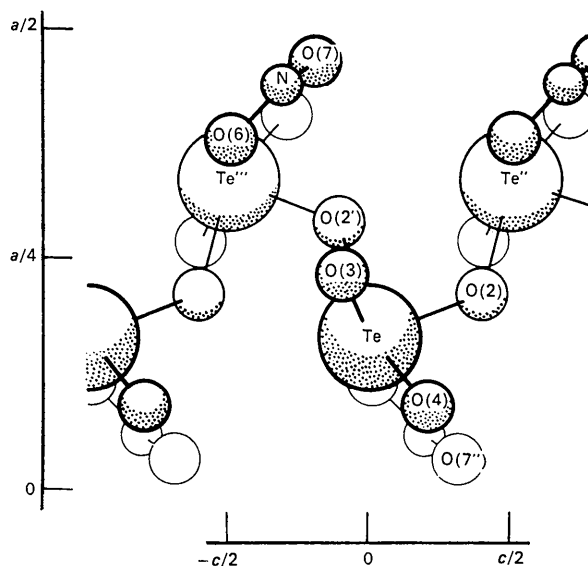


Fig. 2. The section of one layer between mirror planes projected along the  $b$  axis.

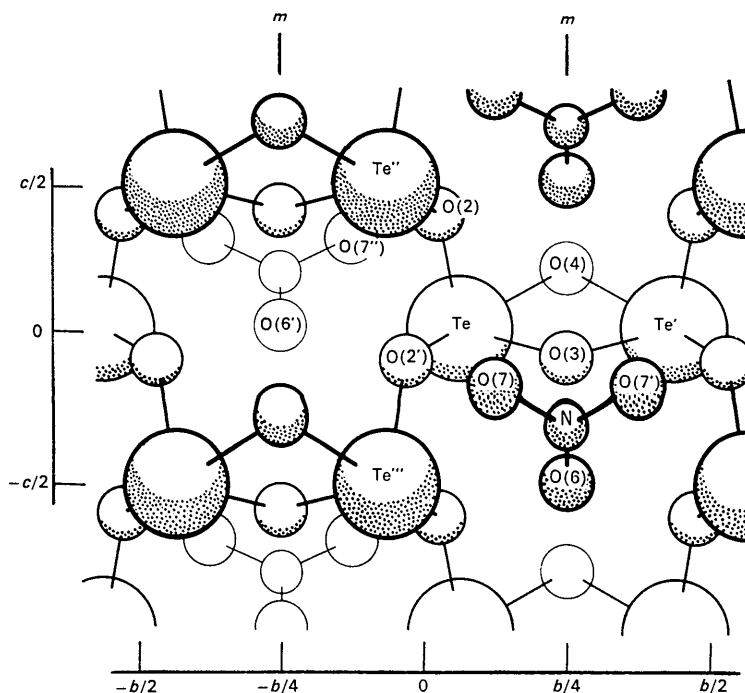


Fig. 1. A section of one layer projected along the  $a$  axis.

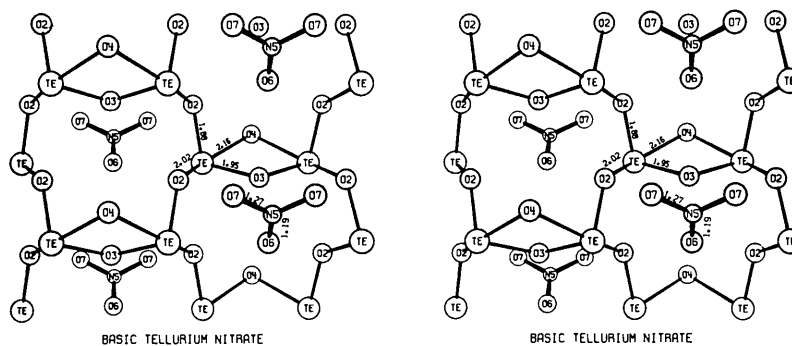


Fig. 3. Stereoscopic drawing of a section of one layer as viewed from the  $a$  axis.

Ridge) for the stereoscopic drawing (Fig. 3) and slide used at the Gatlinburg ACA meeting, and to Professor J. O. Edwards (Brown) who suggested this problem.

This work was supported in part by the Office of Naval Research and in part by the Advanced Research Projects Agency.

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## Die Struktur von $\text{Na}_2\text{H}_2\text{SiO}_4 \cdot 4\text{H}_2\text{O}$

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(Eingegangen am 14. Januar 1966)

$\text{Na}_2\text{H}_2\text{SiO}_4 \cdot 4\text{H}_2\text{O}$  crystallizes in space group  $P\bar{1}$ , with the cell dimensions

$$\begin{aligned} a &= 7.96, & b &= 9.61, & c &= 6.67 \text{ \AA}; \\ \alpha &= 70.1, & \beta &= 104.3, & \gamma &= 122.5^\circ. \end{aligned}$$

The structure was determined by direct methods. Isolated  $\text{SiO}_4$  tetrahedra are linked by hydrogen bonds to form layers which are separated by layers of Na and  $\text{H}_2\text{O}$ . The problem of the condensation of single  $\text{SiO}_4$  tetrahedra to pairs,  $\text{Si}_2\text{O}_7$ , during solution is discussed.

#### Einleitung

Thilo, Wieker & Stade (1965) entwickelten eine Methode zur Bestimmung des Kondensationsgrades löslicher Silicate. Als Mass für den Kondensationsgrad nehmen sie die Geschwindigkeit, mit der sich Silicomolybdat bildet, wenn ein in Säuren leicht lösliches Silicat oder eine frisch hergestellte Silicat-Lösung in eine

saure Lösung von Ammoniummolybdat gebracht werden. Die Temperatur, bei der üblicherweise gearbeitet wird, ist  $+25^\circ\text{C}$ . Einige Silicate, vor allem das  $\text{Na}_2\text{O} \cdot \text{SiO}_2 \cdot 5\text{H}_2\text{O}$  zeigen jedoch bei dieser Reaktion ein bemerkenswertes Verhalten: Löst man das  $\text{Na}_2\text{O} \cdot \text{SiO}_2 \cdot 5\text{H}_2\text{O}$  bei  $+2^\circ\text{C}$ , so verhält es sich bei der Molybdat-Reaktion wie ein Monosilicat, löst man es bei  $+25^\circ\text{C}$ , so verhält es sich wie eine Mischung aus Mono- und